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High degree of polarization of the near-band-edge photoluminescence in ZnO nanowires

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Abstract

We investigated the polarization dependence of the near-band-edge photoluminescence in ZnO strain-free nanowires grown by vapor phase technique. The emission is polarized perpendicular to the nanowire axis with a large polarization ratio (as high as 0.84 at 4.2 K and 0.63 at 300 K). The observed polarization ratio is explained in terms of selection rules for excitonic transitions derived from the k-p theory for ZnO. The temperature dependence of the polarization ratio evidences a gradual activation of the X_C excitonic transition.

PACS: 78.55.Cr, 77.22.Ej, 81.07.Gf.

Keywords: zinc oxide, nanowire, photoluminescence, polarization

Introduction

One-dimensional nanoscale semiconductors have recently attracted considerable attention as promising candidates for innovative device applications. Their high surface to volume ratio can be exploited for the development of a new generation of chemical and biological sensors [1-3]. The wide bandgap (3.37 eV) of ZnO associated with its large exciton binding energy (60 meV) also makes it one of the most promising materials for photonic devices, such as light-emitting diodes [4] and lasers [5]. Thanks to the spatial separation of photogenerated carriers, UV photodetectors with a very high photoconductive gain based on ZnO nanowires (NWs) have been demonstrated [6]. It has been shown that the photodetection properties of ZnO NWs depend on the light polarization [7].

The photoluminescence of ZnO is typically composed of a near-band-edge (NBE) peak due to excitonic recombination and of a broad emission band in the visible range related to deep defect states [8-10]. The polarization properties of the luminescence of ZnO have been studied in bulk crystals [11-15]. However, these studies provided no theoretical explanation of the polarization behavior, especially of its temperature dependence. In

the specific case of NWs, several studies have been carried out, but they were focused on the interpretation of the different behavior of defect and NBE luminescence [16,17]. As shown in other semiconductor NW systems [18], the polarization dependence in NWs results from two competitive phenomena: bulk crystal symmetry (imposing polarization perpendicular to *c*-axis) [19] and dielectric contrast in thin NWs (privileging polarization parallel to the NW axis) [7,20-23].

In this work, we have studied the polarization-resolved microphotoluminescence (μ -PL) of ZnO nanowires. We measured the polarization dependence of the NBE luminescence for temperature from 4.2 to 300 K. The experimental results are interpreted in the framework of the k-p model, allowing for the evaluation of the polarization ratio for each exciton type in bulk ZnO. The temperature dependence of the polarization ratio evidences a gradual activation of the X_C excitonic transition.

Experimental details

ZnO NWs are prepared by means of vapor transport process, in which the source material is vaporized and transported by a gas carrier towards the substrates where it condenses [24]. The experimental setup consists of a furnace capable to reach temperatures needed for oxide evaporation, a vacuum-sealed alumina tube connected to a vacuum pump, an automated valve, and

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a mass flow meter to control pressure and carrier flux. Adjusting the deposition conditions such as temperature of evaporation and carrier gas composition and flux, one-dimensional nanostructures can be obtained.

Platinum catalyst particles are firstly dispersed onto silicon substrates by DC magnetron sputtering at a working pressure of 5×10^{-3} mbar and 50 W applied power. The source material is positioned at the middle of the alumina tube and evaporated at a temperature of 1,370°C at a pressure of 100 mbar. The platinum catalyzed substrates are placed onto an alumina holder and positioned inside the tube in an area corresponding to a temperature $T = 660^\circ\text{C}$. Furnace heating from room temperature to 1,370°C lasts 1.5 h. During furnace heating and cooling, a reverse Ar gas flow (from the substrates to the powder) is applied to avoid uncontrolled mass deposition under transient conditions. Once the desired temperature is reached, the deposition conditions are kept for 15 min, and afterwards, the furnace is cooled down to room temperature.

As seen in Figure 1a, transmission electron microscopy (TEM) analysis shows that the NWs are single

crystalline ZnO with a wurtzite structure. The NW axis is oriented along the [0001] direction, and the structure is free from extended defects. The high-resolution TEM images evidence well-defined lateral sidewalls, parallel to the wire growth direction. No impurities or precipitates have been detected within the accuracy of the energy dispersive X-ray spectroscopy performed in the TEM. The NW morphology was observed by scanning electron microscopy (SEM). A typical top-view SEM image of the as-grown NW ensemble is shown in Figure 1b. The SEM analyses show that the NWs do not have a specific direction with respect to the substrate. The diameter is dispersed in the range of 20 to 100 nm and the length is in the range of 500 nm to 5 μm .

Results and discussions

For μ -PL studies, single NWs were detached by ultrasound bath from their substrates and dispersed in ethanol on Si substrates patterned with alignment marks. The surface density of NWs is controlled by dispersion in the range of 1 to 5×10^6 NWs/cm², which is low enough to avoid simultaneous optical excitation of several wires with different orientations. The dispersed NWs do not show any bending and are free of strain. Polarization-resolved μ -PL experiments have been performed in the temperature interval of 4.2 to 300 K. The samples were cooled down in a continuous-flow liquid He cryostat and excited by means of a frequency-doubled continuous-wave Ar⁺⁺ ion laser at 244 nm. The laser was focused on the substrate surface in a spot with a diameter of 3 μm by means of a UV microscope objective with 0.4 numerical aperture. The excitation power was set in the range of 10 to 50 μW . The sample was imaged through a UV-sensitive camera in order to visualize the luminescence spot and to locate the NW with respect to the alignment marks. μ -PL spectra were measured using a Jobin Yvon HR460 spectrometer (Horiba Ltd., Tokyo, Japan) with a 600- or 1,800-grooves/mm grating and a charge-coupled device camera. The energy resolution of the setup during these experiments is around 1 meV. In order to analyze the polarization of the single NW emission, a linear polarizer was placed at the entrance of the spectrometer. For each individual NW, a series of spectra was collected at different angles of the polarizer axis, which was varied over the whole interval 0° to 360° with a 15° step. The orientation of the NW with respect to the polarizer axis, as well as its isolation from other dispersed nanowires, has been assessed by SEM measurements performed after the optical characterization. The experiment was carried out on ten NWs, yielding a good reproducibility.

Using the alignment marks as a reference frame, we identified the polarizer angles corresponding to π - (light with the electric field $E \perp c$ -axis) and σ - ($E // c$ -axis)

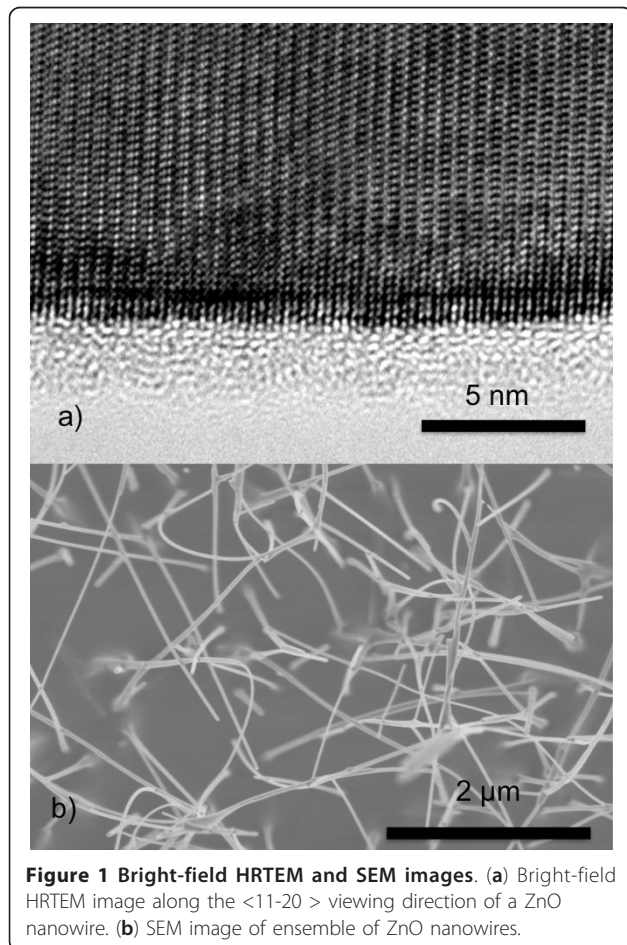


Figure 1 Bright-field HRTEM and SEM images. (a) Bright-field HRTEM image along the $\langle 11\bar{2}0 \rangle$ viewing direction of a ZnO nanowire. (b) SEM image of ensemble of ZnO nanowires.

polarizations for each wire. The polarization ratio is defined as:

$$P = (I_{\pi} - I_{\sigma}) / (I_{\pi} + I_{\sigma}) \quad (1)$$

where I_{π} and I_{σ} are the integrals of the PL intensity for the π and σ polarizations, respectively.

The polarization of the NBE emission is related to the selection rules for the excitonic transitions, which can be deduced from the k-p theory [19]. The polarization ratio of the three exciton types in strain-free crystal can be expressed as a function of the band parameters of ZnO. The interband momentum-matrix elements $M_{\sigma,\pi}^2$ proportional to the PL intensity for σ and π polarization are reported in the Table 1. It shows that the X_A exciton (formed by an electron bound to a heavy hole) is purely polarized perpendicular to the c -axis due to the selection rules in wurtzite crystal. The X_B exciton (an electron bound to a light hole) is strongly polarized perpendicular to the c -axis, whereas the X_C exciton (an electron bound to a split-off hole) is strongly polarized parallel to the c -axis. The X_C exciton has a much higher energy than the X_A one (energy difference ΔE_{CA} between the X_C and X_A is 48 meV). Therefore, the NBE photoluminescence is dominated by the lower-energy X_A exciton and, in consequence, is expected to be strongly polarized perpendicular to the c -axis even at room temperature ($k_B T \approx 25$ meV $<$ ΔE_{CA}). It should be noted that many-particle processes can potentially influence the emission polarization. In the polarization analyses, we approximate the NBE emission as originating solely from the X_A bound and free excitons and we neglect the effect of phonon replicas, which are one order of magnitude weaker than the main peak and which could not be detected in single nanowire spectra.

The PL spectrum of the NW ensemble collected at 4.2 K is reported in Figure 2a. It presents a broad NBE emission peaked at 3.357 eV consisting of different contributions from the bound states of the X_A exciton and possibly a contribution of X_B exciton, which cannot be separately resolved. In addition, a weak shoulder is observed at high energy (3.41 eV) which is related to the X_C exciton. The μ -PL spectra of two single NWs recorded with a spectral resolution of 800 μ eV are

Table 1 Normalized interband squared momentum-matrix element $M_{\sigma,\pi}^2$

Exciton type	$E // c$ -axis [ZnO value]	$E \perp c$ -axis [ZnO value]
X_A	0	0.5
X_B	0.0057	0.4965
X_C	0.9929	0.035

For polarization along the c -axis and perpendicular to the c -axis for excitons X_A , X_B , and X_C for ZnO crystal.

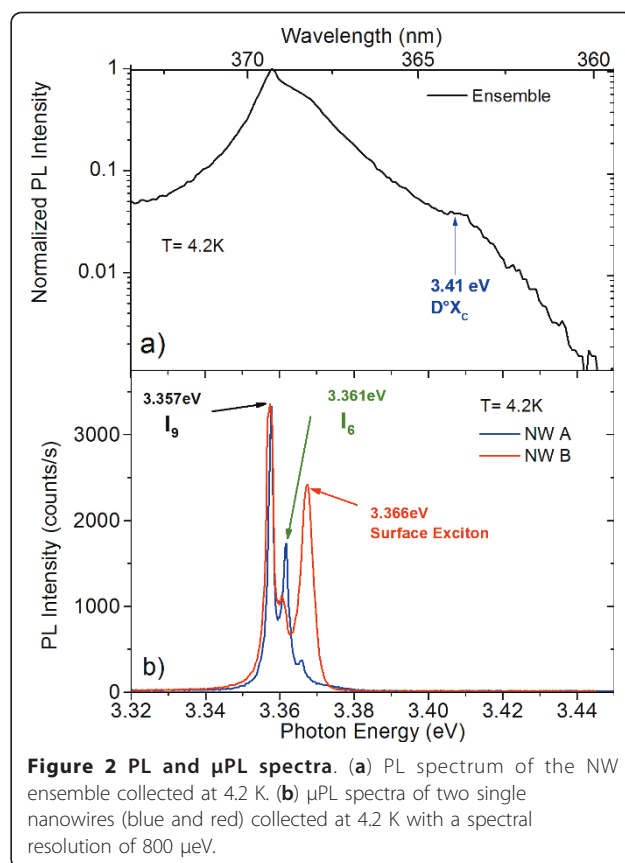
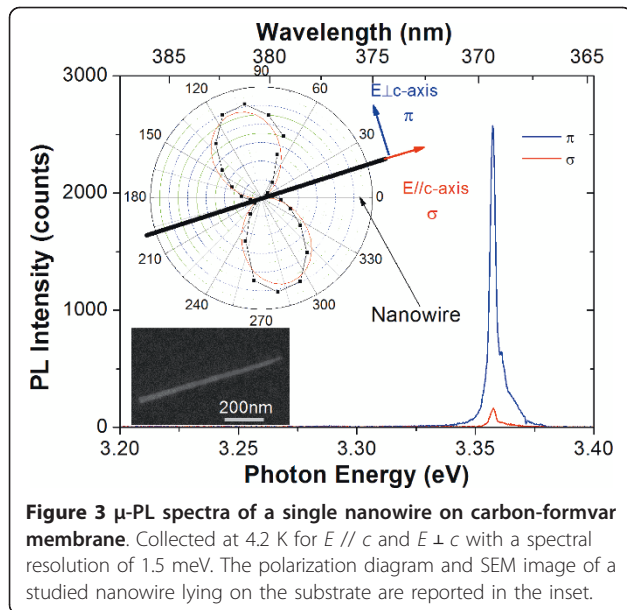


Figure 2 PL and μ PL spectra. (a) PL spectrum of the NW ensemble collected at 4.2 K. (b) μ PL spectra of two single nanowires (blue and red) collected at 4.2 K with a spectral resolution of 800 μ eV.

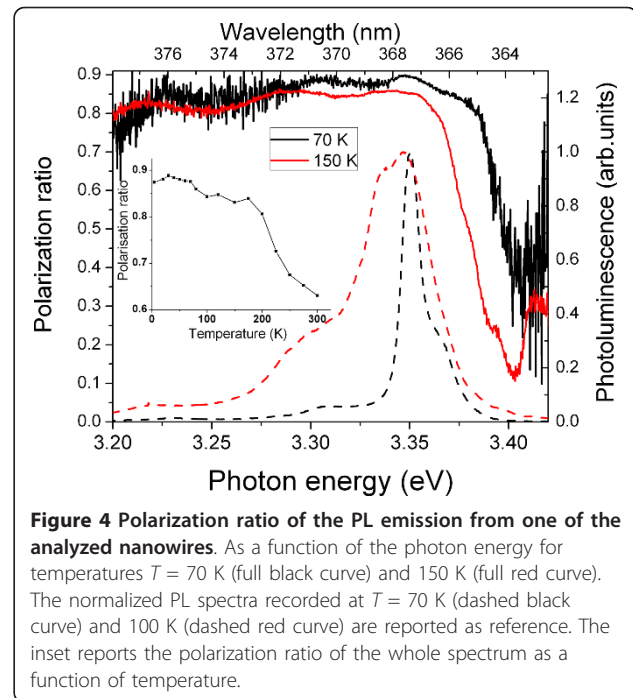
reported in Figure 2b. All spectra exhibit three narrow peaks with linewidth as low as 1.5 meV. These peaks can be attributed to the different bound states of X_A exciton. The two predominant peaks at 3.357 and 3.361 eV are attributed to the I_9 and I_6 lines related to the neutral donor-bound exciton $D^{\circ}X_A$, respectively, bound to Al and In [25]. Alternatively, as studied by Meyer et al. [26], these lines could be related to the neutral donor-bound exciton $D^{\circ}X_A$ and $D^{\circ}X_B$. The latter interpretation is less probable since the relative intensity of the peaks does not match with the expected population of the corresponding excitonic states at low temperature. In the following analyses, we use the first attribution; however, the conclusions also remain valid for the second one. At higher energy (3.366 eV), we notice a third peak related to a surface bound X_A exciton [27,28]. The intensity of this peak varies from wire to wire due to the NW size dispersion.

Typical μ -PL spectra recorded at $T = 4.2$ K for σ - and π -polarizations are reported in Figure 3. By changing the polarization from σ to π , we observe that the spectral shape remains the same within the experimental accuracy, but the PL intensity integrated on the entire spectrum varies of about a factor of 12. This large contrast corresponds to a polarization ratio as



high as 0.85. The statistics over ten NWs yields an average polarization ratio of 0.84 with a standard deviation of 0.05. The dependence of the PL intensity on the angle of polarization can be well fitted by a cosine-squared law $I \approx \cos^2(\pi/2 - \theta)$, where θ is the angle between the analyzer and the NW axis determined from the SEM analyses. The maximum luminescence intensity is obtained when the analyzer is perpendicular to the c -axis of the NW (π -polarization). From the polarization selection rules, an even higher polarization ratio of 0.98 is expected. The difference between the experimental observation and the theoretical prediction can possibly be explained by a partial depolarization due to the diffraction from the NW of the luminescence exiting the NW extremities. In addition, the dielectric contrast between the NW and its environment, and the elongated shape of ZnO NWs with small diameter (<80 nm), should favor the emission of light polarized parallel to the NW axis [20,21]. However, this effect cannot compete with the high anisotropy of the emission polarization. The polarization of the ZnO luminescence perpendicular to the NW axis is dictated by the excitonic selection rules.

The temperature dependence of the polarization ratio integrated on the whole spectrum is reported in the inset to the Figure 4. The polarization ratio decreases when the temperature increases from $P = 0.85$ at $T = 4$ K to $P = 0.63$ at $T = 300$ K. This effect is due to the progressive thermal activation of higher energy excitons, in particular of the X_C having a different symmetry. However, the X_C population remains weak even at room



temperature, which explains a high polarization ratio (above 0.63) over the whole interval $T = 4$ to 300 K.

It is difficult to observe the gradual activation of the X_C emission directly from the μ -PL spectra because of the very weak signal associated with this transition. However, the X_C luminescence can be evidenced by analyzing the energy-dependent polarization ratio $P(E)$:

$$P(E) = (I(E)_\pi - I(E)_\sigma) / (I(E)_\pi + I(E)_\sigma) \quad (2)$$

at different temperatures, where $I(E)_p$ is the PL intensity at energy E in the p polarization. Figure 4 reports the $P(E)$ for 70 and 150 K. (The temperature range is restricted within 70 to 150 K due to the extremely low signal above 3.38 eV at low temperatures and to the decrease of the overall luminescence intensity at high temperature). For the energy region between 3.28 and 3.38 eV, the signal arises from the X_A and thermally activated X_B excitonic transitions. Therefore, the polarization ratio remains high (>0.85) in this interval and is nearly temperature independent. At higher energy, around 3.41 eV, the polarization ratio decreases in correspondence of the emission of the X_C exciton. It should be noted that in spite of the weak signal in the spectral range corresponding to the X_C emission, the signal-to-noise ratio is about 20 at 3.40 eV. Therefore, the maximum possible error on the polarization ratio induced by the noise is less than 0.1. With increasing temperature, the dip in the $P(E)$ dependence is amplified and

progressively shifts towards lower energy. This behavior reflects the progressive thermal activation of the X_C excitonic emission and the ZnO bandgap reduction described by the Varshni law [29].

Conclusions

In conclusion, we have studied the optical properties of ZnO nanowires grown by evaporation technique. Nanowires have defect-free single crystalline structure as shown by high-resolutions TEM (HRTEM) analysis. The nanowires are characterized by an intense photoluminescence with a spectral broadening below 2 meV. We have investigated the polarization dependence of the near-band-edge photoluminescence in ZnO strain-free nanowires. They exhibit a polarization ratio as high as 0.84. We show that these observations are consistent with the k-p theory and with the exciton selection rules. In particular, the weak dependence of the integrated polarization ratio P is a consequence of the large energy difference between X_A and X_C excitons. However, the analysis of the energy-resolved polarization ratio $P(E)$ at different temperatures allows for the observation of the progressive activation of the X_C exciton.

Abbreviations

NBE: near band edge; NW: nanowire; PL: photoluminescence; μ -PL: microphotoluminescence; SEM: scanning electron microscopy; TEM: transmission electron microscopy.

Acknowledgements

This work was supported by the French ANR agency under the programs ANR-08-NANO-031 BoNaFo and ANR-08-BLAN-0179 NanoPhotoNit.

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Authors' contributions

GJ carried out the μ -PL measurements and data analysis, performed k-p analysis, and wrote the manuscript. LR and ADLB participated in the μ -PL measurements. LR, MT, and FHJ participated in the data analysis and to the interpretation of the results. CB and EC grew the sample. MF performed the TEM analysis. All authors read and approved the final manuscript.

Competing interests

The authors declare that they have no competing interests.

Received: 13 April 2011 Accepted: 19 August 2011
Published: 19 August 2011

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doi:10.1186/1556-276X-6-501

Cite this article as: Jacopin *et al.*: High degree of polarization of the near-band-edge photoluminescence in ZnO nanowires. *Nanoscale Research Letters* 2011 **6**:501.

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